Thermal analysis of formation of nano-crystalline BaTiO₃ using Ba(NO₃)₂ and TiO₂

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Abstract

The reaction of Ba(NO₃)₂ with TiO₂ was studied by thermogravimetric (TG) and differential scanning calorimetric (DSC) techniques up to 1000 °C and in nitrogen atmosphere. It was found that the formation of BaTiO₃ takes place above 600 °C and that precursor mixing time and heating rate have no effect on the reaction temperature. BaTiO₃ powder was prepared by calcination of Ba(NO₃)₂ and TiO₂ precursor mixture at 800 °C for 8 h. X-ray diffraction analysis of the synthesized BaTiO₃ confirmed the formation of tetragonal phase with lattice parameters a = 3.9950±0.0003 Å and c = 4.0318±0.0004 Å. Thermal analysis of the synthesized BaTiO₃ powder showed weight loss within temperature range 40–1000 °C of only 0.40%. This small amount of weight loss was connected with some impurity phase, and identified as BaCO₃ using Fourier transform infrared (FTIR) technique.

Keywords: synthesis, barium titanate, Ba(NO₃)₂ precursor, thermal analysis, XRD, FTIR

I. Introduction

Performance of BaTiO₃ ceramics significantly depends on the microstructure of the sintered body, which is influenced by characteristics of the starting BaTiO₃ powder. Therefore, much attention has been focused on the synthesis of high quality BaTiO₃ nano-particles and different techniques have been used: sonochemical synthesis [1], sol-gel [2], hydrothermal [3], solvothermal [4], chemical co-precipitation [5] and supercritical fluid [6]. These methods are not cost-effective in mass production of BaTiO₃ powders. Thus, preparation of BaTiO₃ powders, using solid state reaction synthesis is regarded as one of the most practical and economical ways [7]. However, in the conventional solid phase methods, BaTiO₃ powders were usually produced via the reaction between the mechanically mixed BaCO₃ or BaO and TiO₂ powders followed by calcination at above 1000 °C [8]. Unfortunately, the reactions at such high temperatures are beyond control and the resultant powders often have large particle size and wide particle size distribution. Therefore, for industrial applications, it is still desirable to develop an alternative low-temperature solid state method for the preparation of nano-crystalline BaTiO₃ powders at low cost.

Researchers have tried to synthesize BaTiO₃ powders at lower temperatures by replacing BaCO₃ by other compounds such as Ba(OH)₂, Ba(COO₂), Ba(NO₃)₂ etc. [9]. Thus, it was reported that the reaction of Ba(NO₃)₂ with TiO₂ (anatase) yielded the single phase BaTiO₃ at much lower temperatures, around 600 °C [10–12], but this method has not been studied in details like a synthesis of BaTiO₃ using BaCO₃ and TiO₂. Therefore, the aim of this work is to study the formation of BaTiO₃ from Ba(NO₃)₂ and TiO₂, using thermogravimetric (TG) and differential scanning calorimeter (DSC) analysis. Further characterization of the prepared BaTiO₃ powder was done using thermal analysis (TG and DSC), X-ray diffraction (XRD) and Fourier Transform Infra-red (FTIR) techniques.

II. Experimental

2.1. Sample preparation

BaTiO₃ powder was prepared by using Ba(NO₃)₂ (purity 99%, Qualikens, USA) and titanium dioxide (rutile TiO₂ purity 99.5%, Alfa Aesar, UK) as the starting
materials. Equi-molar amounts of Ba(NO₃)₂ and TiO₂ powders were crushed in agate mortar for 3 hours. The mixture was transferred to an alumina crucible and calcined in a furnace (Matrix, India) at 800 °C for 8 h with a heating rate of 10 °C/min and after that the furnace was cooled down to room temperature.

2.2. Characterization techniques

Thermal analysis (TG and DSC) of the reactants, precursor mixture of reactants and synthesized powders were carried out by using simultaneous TG-DSC (Mettler Toledo, Germany) thermal analyzer in the temperature range 30–1000 °C with heating rates of 10 and 20 °C/min in nitrogen atmosphere. A preliminary study on compound formation and lattice parameters was carried out by using X-ray diffraction (XRD) technique with an X-ray powder diffract meter (Miniflex, Rigaku). Fourier transform infrared (FTIR) spectra of the sample were recorded by using Fourier Transform Infra-red Spectrometer (Shimadzu, Model DF 803) in the wave number range 400–4000 cm⁻¹.

III. Results and discussion

3.1. Thermal analysis of TiO₂ and Ba(NO₃)₂

The TG and DSC curves of TiO₂ powder with a heating rate of 10 °C/min in nitrogen atmosphere are shown in Figs. 1a and 1b, respectively. No weight change and peaks have been observed in the TG and DSC curves of TiO₂ powder in the temperature range 30–1000 °C.

The TG and DSC curves of Ba(NO₃)₂ are shown in Figs. 2a and 2b, respectively. The TG curve shows a single step weight loss (approximately 41%) in the temperature range 550–725 °C. It was reported [13] that anhydrous bulk barium nitrate decomposes over the temperature range 500–700 °C, with the main decomposition taking place at 630 °C. Slightly higher decomposition temperature range observed for nitrate used in this work may be due to difference in the particle size. The observed weight loss is in good agreement with the theoretical weight loss calculated according to the reaction (1):

$$\text{Ba(NO}_3\text{)}_2 \rightarrow \text{BaO} + 2 \text{NO}_2 + 0.5 \text{O}_2$$ (1)

The DSC curve of Ba(NO₃)₂ shows two peaks at 587 °C and 708 °C. The first peak is attributed to the melting of Ba(NO₃)₂ whereas the second peak to the decomposition of Ba(NO₃)₂ to BaO according to the reaction (1). The observed melting temperature is approximately the same as reported value of 592 °C [14].

3.2. Thermal analysis of Ba(NO₃)₂/TiO₂ mixture

Figures 3 and 4 show the TG and DSC curves, respectively, of the mixture of Ba(NO₃)₂ and TiO₂ obtained upon heating with two different rates (10 and 20 °C/min) and for two different mixing times (1 and 3 hours). There is no significant change in the nature of TG curves with variations in the heating rate and mixing time. All three TG curves show a single step weight loss (approximately 31%) between 550–625 °C. The formation of BaTiO₃ perovskite phase, in single step accompanied by the release of NO₂ and O₂ gases, takes place according to the following equation:

$$\text{Ba(NO}_3\text{)}_2 + \text{TiO}_2 \rightarrow \text{BaTiO}_3 + 2 \text{NO}_2 + 0.5 \text{O}_2$$ (2)
The weight loss of 31% observed experimentally is in agreement with the theoretical weight loss calculated according to the reaction (2).

DSC curves show only one endothermic peak between 610–650 °C. Since this peak is above the melting temperature of Ba(NO$_3$)$_2$, it is attributed to the liquid-solid reaction between Ba(NO$_3$)$_2$ and TiO$_2$ during the formation of BaTiO$_3$. Observations of a single peak in the DSC curve contradict to the results reported by other workers. Tagawa et al. [10] have observed two endothermic peaks in the DTA curve. They attributed the first peak, at ~600 °C, to the melting of Ba(NO$_3$)$_2$ and the second one, at ~650 °C, to the reaction of Ba(NO$_3$)$_2$ with TiO$_2$. They concluded that the reaction occurs in the liquid-solid phase after the melting of Ba(NO$_3$)$_2$ at high heating rates (10 and 20 °C/min). Othman et al. [12] showed that the reaction between Ba(NO$_3$)$_2$ and TiO$_2$ proceeds during heating up in three consecutive stages, characterized by three endothermic peaks at 605, 633.7 and 675.2 °C in the DTA curve. The first peak (at 605 °C) might be attributed to the solid-solid reaction between Ba(NO$_3$)$_2$ and TiO$_2$ powders. The second peak (at 633.7 °C) is due to the reaction between the partially melted Ba(NO$_3$)$_2$ and TiO$_2$. The last endothermic peak (at 675.2 °C) is due to the complete reaction of the molten Ba(NO$_3$)$_2$ with the residual TiO$_2$.

3.3. Characterization of synthesized BaTiO$_3$ powder

Figure 5 shows TG-DSC curves of the synthesized BaTiO$_3$ powder, obtained after calcination at 800 °C for 8 h at a heating rate of 10 °C/min in nitrogen atmosphere. The TG curve depicts only 0.4% weight loss below 800 °C which could be attributed to the decomposition of a small amount of an impurity phase. There is no distinct peak in the DSC curve of the obtained powder which reflects high purity and thermal stability of the synthesized material.

The X-ray diffraction (XRD) pattern of synthesized BaTiO$_3$ powder is shown in Fig. 6 and confirms the formation of BaTiO$_3$. The crystal structure of BaTiO$_3$ powder is tetragonal (JCPDS No. 15-0780) because both 200 and 002 peaks (around 2θ = 45°) are present in the XRD pattern (see the inset of Fig. 6). The lattice parameters obtained using software “Cell” are $a = 3.9950 ± 0.0003$ Å and $c = 4.0318 ± 0.0004$ Å. The value of the tetragonality calculated using these values of lattice parameters is 1.0092 which is slightly smaller than standard value 1.0110. This may be due to the lower synthesis temperature used. It is reported that tetragonality of BaTiO$_3$ is a function of temperature and standard value of tetragonality can be achieved after sintering at temperature >1300 °C [15].

The average crystallite size of the synthesized BaTiO$_3$ powder was calculated from the X-ray line broadening of the 110 peak using Scherrer’s formula:

$$D = \frac{0.9 \lambda}{\beta \cos \theta_{\text{max}}}$$  (3)
where $D$ is the average grain size, $\lambda = 1.541 \text{ Å} \ (\text{X-ray wavelength})$, and $\beta$ is the width of the diffraction peak at half maximum for the diffraction angle $2\theta$. The average particle size is found to be 44 nm.

According to the reported literatures, BaCO$_3$ is the most common impurity of the BaTiO$_3$ powder prepared by any method. FTIR was found to be the most sensitive technique for the detection of BaCO$_3$. On the basis of FTIR analysis, BaCO$_3$ concentration down to 0.6% has been estimated. In order to check the purity of the obtained BaTiO$_3$ powder, FTIR spectra of the sample is recorded and shown in Fig. 7. The crystallization of barium titanate was confirmed by the broad band centered at 540 cm$^{-1}$, which is typical of the Ti–O stretching vibrations in BaTiO$_3$, and also by the second broad band observed at the end of the spectra, below 500 cm$^{-1}$, corresponding to the Ti–O bending vibrations [16]. The very small absorption band at 1440 cm$^{-1}$ can be interpreted as C–O vibration due to extremely small unavoidable traces of carbonate [17]. This result is contradictory to the result obtained by XRD (Fig. 6 in which no peak of BaCO$_3$ is observed), but is in agreement with results obtained by TG curve (Fig. 5). From the TG it was found that amount of impurity phase BaCO$_3$ is approximately 0.4% and as it could be expected XRD is unable to detect such small amount of BaCO$_3$ impurity phase.

IV. Conclusions

The BaTiO$_3$ powder was synthesized from Ba(NO$_3$)$_2$ and TiO$_2$ powder mixture with 1:1 molar ratio. The TG-DSC results showed that the reaction between Ba(NO$_3$)$_2$ and TiO$_2$ proceeds during heating up in single stage (above 600 °C) between molten Ba(NO$_3$)$_2$ with TiO$_2$ (liquid-solid reaction). The TG-DSC results showed that the formation of BaTiO$_3$ phase from Ba(NO$_3$)$_2$ and TiO$_2$ powder mixture takes place before Ba(NO$_3$)$_2$ decomposes to BaO. X-ray diffraction pattern of the synthesized BaTiO$_3$ obtained after calcining mixture of BaTiO$_3$ and TiO$_2$ at 800 °C for 8 h confirmed the formation of tetragonal phase with lattice parameters $a = 3.9950{\pm}0.0003 \text{ Å}$ and $c = 4.0318{\pm}0.0004 \text{ Å}$. The obtained powder has crystallite size of 44 nm. The purity of the obtained powder has been checked by thermal characterization (TG and DSC). Only 0.40% weight loss has been detected within temperature range 40–1000 °C. This small amount of impurity phase was identified as BaCO$_3$ using Fourier transform infra-red (FTIR) technique.

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